

SOME EFFECTS OF R(R=Li,Na AND K) IONS ON CRYSTAL GROWTH OF $RNd(WO_4)_2$
CRYSTALS FROM TERNARY SYSTEMS $Nd_2O_3-WO_3-R_2CO_3$

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$RNd(WO_4)_2$ (R=Li,Na and K) crystals were grown from high-temperature solutions of ternary systems $Nd_2O_3-WO_3-R_2CO_3$ by slow cooling. Grown $RNd(WO_4)_2$ (R=Li and K) crystals were generally bigger than $NaNd(WO_4)_2$ crystals. These crystals, whose shapes were octahedron or decahedron, were reddish purple and mostly transparent. Among these tungstate crystals, $NaNd(WO_4)_2$ crystals had the highest melting point of $1235\pm 5^\circ C$.

$LiNd(WO_4)_2$ ¹⁾, $NaNd(WO_4)_2$ ²⁾ and $KNd(WO_4)_2$ (high-temperature polymorph)³⁾ crystals have a common crystal structure which is similar to that of tetragonal scheelite ($CaWO_4$) under atmospheric pressure. The low-temperature polymorph of $KNd(WO_4)_2$ crystals belongs to the monoclinic system.⁴⁾ These crystals are of physical interest for their optical⁵⁾ and magnetic⁶⁾ properties.

Some attempts were made to grow these crystals by the flux method (Li ¹⁾, Na ⁷⁻⁹⁾ and K ¹⁰⁾), the solid state reaction (Li ¹¹⁾, Na ^{5,12,13)} and K ¹⁴⁾) and the other methods.¹⁵⁾

However, a study on the crystal growth of these three kinds of crystals from high-temperature solutions under the same growth conditions has not been reported as yet.

The authors have reported not only the crystal growth of $NaLn(WO_4)_2$ ($Ln=La,Nd$ and Gd) crystals from the ternary systems $Ln_2O_3-WO_3-Na_2CO_3$ ¹⁶⁾ but also that of $Na(La,Nd)(WO_4)_2$ solid solution crystals.¹⁷⁾

The main purpose of the present study is to investigate the effects of R ions on crystal growth of $RNd(WO_4)_2$ crystals. Therefore, both $LiNd(WO_4)_2$ and $KNd(WO_4)_2$ crystals were newly synthesized, and their colors, transparencies, sizes, habits, lattice parameters, densities and melting points were examined in addition to $NaNd(WO_4)_2$ crystals which were previously studied.¹⁶⁾

The chemicals used for the preparation of starting materials were Nd_2O_3 (99.9%), WO_3 (reagent grade) and R_2CO_3 (reagent grade). The ternary systems were classified into the following three groups.

- (i) NWL ternary system : $Nd_2O_3-WO_3-Li_2CO_3$
- (ii) NWN ternary system : $Nd_2O_3-WO_3-Na_2CO_3$
- (iii) NWK ternary system : $Nd_2O_3-WO_3-K_2CO_3$

The compositions of these ternary systems were chosen with reference to the optimum ones for the crystal growth of $NaNd(WO_4)_2$ crystals.¹⁶⁾

The batch (52-59 g) was put into a 30 cm³ platinum crucible. The crucible was placed in an electric muffle furnace. The temperature conditions for crystal growth were as follows.¹⁶⁾

Soaking temperature : 1150°C, Soaking period : 10 hours

Cooling rate : 5°C/hr, Cooling range : 1150-500°C

After a run, the crystals were separated from the solidified fluxes with hot water.

Table 1 represents Run No., starting compositions and the maximum sizes of grown crystals. Excepting the losses of CO₂ gas, the losses of fluxed melts in these growth experiments due to evaporation at high-temperatures were less than 2 % in weight at the ends of each run. Therefore, it can be considered that there was little effect of evaporation of fluxed melt on the crystallization.

Grown crystals, which were reddish purple, were mainly attached to the bottom or wall of the crucible in each case, regardless of the difference in species of R ions. These crystals were identified by comparison of their X-ray powder patterns

with published data of RNd(WO₄)₂ crystals (Li^{1,11}, Na^{2,8,12,13} and K^{3,10}), respectively. As a result, KNd(WO₄)₂ crystals were found to be tetragonal high-temperature polymorph.

As listed in Table 1, grown RNd(WO₄)₂ (R=Li and K) crystals were generally bigger than NaNd(WO₄)₂ crystals. The respective biggest RNd(WO₄)₂ crystals were obtained from the batches containing 5 mol% of Nd₂O₃. Therefore, as can be seen from Table 1, NWL-1, NWN-1 and NWK-1 compositions were optimum for the crystal growth of respective RNd(WO₄)₂ crystals. On the basis of the assumption that these compositions were pseudobinary systems consisting of solute RNd(WO₄)₂-flux R₂W_xO_y, the flux compositions were regarded as R₂W_{1.8}O_{6.4}. These values (x=1.8, y=6.4) indicate that the flux compositions were not far from R₂W₂O₇.

Each example of grown crystals for Li or K is shown in Fig.1 or 2. By measuring the interfacial angles of these crystals, it is shown that the shapes and habits of both LiNd(WO₄)₂ and KNd(WO₄)₂ crystals are classified into the following two groups in the same manner as with those of NaNd(WO₄)₂ crystals.¹⁶⁾

Octahedron : {101} faces

Decahedron : {101}+{001} faces

Table 1 Starting compositions and maximum sizes of grown RNd(WO₄)₂ crystals

Run No.	Starting composition (mol%)	Maximum size of crystal (mm)
(i) — [NWL-1	Nd ₂ O ₃ (5) -WO ₃ (65) -Li ₂ CO ₃ (30)	8x6x5
NWL-2	Nd ₂ O ₃ (3) -WO ₃ (67) -Li ₂ CO ₃ (30)	4x3x3
NWL-3	Nd ₂ O ₃ (2) -WO ₃ (67) -Li ₂ CO ₃ (31)	3x3x2
(ii) — [NWN-1	Nd ₂ O ₃ (5) -WO ₃ (65) -Na ₂ CO ₃ (30)	4x4x3
NWN-2	Nd ₂ O ₃ (3) -WO ₃ (67) -Na ₂ CO ₃ (30)	4x4x3
NWN-3	Nd ₂ O ₃ (2) -WO ₃ (67) -Na ₂ CO ₃ (31)	3x3x2
(iii) — [NWK-1	Nd ₂ O ₃ (5) -WO ₃ (65) -K ₂ CO ₃ (30)	8x6x5
NWK-2	Nd ₂ O ₃ (3) -WO ₃ (67) -K ₂ CO ₃ (30)	7x7x6
NWK-3	Nd ₂ O ₃ (2) -WO ₃ (67) -K ₂ CO ₃ (31)	5x3x3

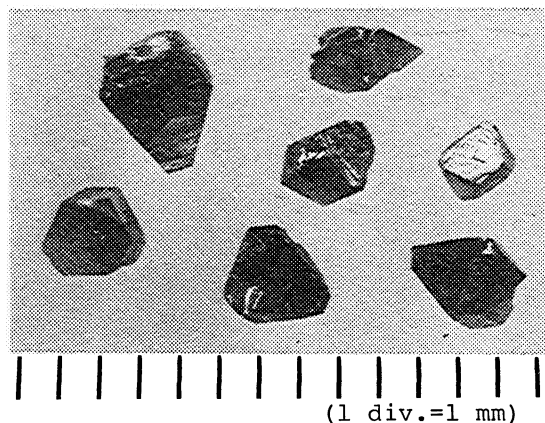


Fig.1 LiNd(WO₄)₂ crystals
(Run No. NWL-2)

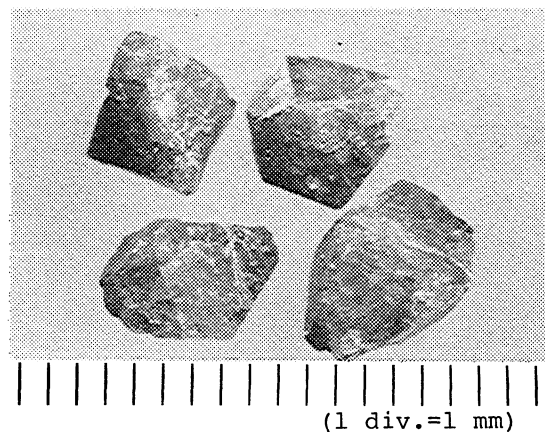


Fig.2 KNd(WO₄)₂ crystals
(Run No. NWK-2)

Table 2 Lattice parameters, unit cell volumes and formula weights of $\text{RNd}(\text{WO}_4)_2$ crystals

Crystal	Lattice parameter		Unit cell volume (\AA^3)	Formula weight
	a (\AA)	c (\AA)		
$\text{LiNd}(\text{WO}_4)_2$	5.261 \pm 0.004	11.43 \pm 0.03	316.4(1.000)	646.9(1.000)
$\text{NaNd}(\text{WO}_4)_2$	5.300 \pm 0.004	11.52 \pm 0.03	323.6(1.023 ¹⁸)	662.9(1.025 ²⁰)
$\text{KNd}(\text{WO}_4)_2$	5.372 \pm 0.004	11.93 \pm 0.03	344.3(1.088 ¹⁹)	679.0(1.050 ²¹)

Therefore, it has been pointed out that the habits of $\text{RNd}(\text{WO}_4)_2$ crystals don't depend upon the difference of R ions. Regardless of the difference of R ions, a large number of $\text{RNd}(\text{WO}_4)_2$ crystals were transparent. According to the observation under a microscope, visible inclusions were rarely found in these crystals. However, as for the machinability, $\text{LiNd}(\text{WO}_4)_2$ and $\text{KNd}(\text{WO}_4)_2$ crystals seemed to be somewhat more brittle than $\text{NaNd}(\text{WO}_4)_2$ crystals.

In Table 2, lattice parameters (a and c) observed by using X-ray data, calculated unit cell volumes and formula weights of $\text{RNd}(\text{WO}_4)_2$ crystals are tabulated. The lattice parameters a and c increased with increase of the ionic radii of R ions in a common tetragonal scheelite structure.

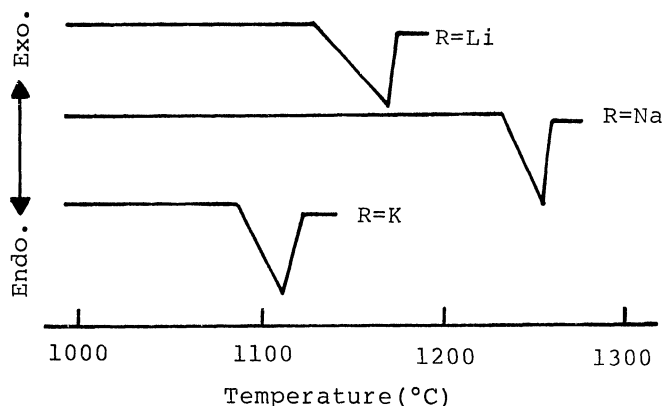
Table 3 gives densities, which were pycnometrically determined at $20.0\pm 0.5^\circ\text{C}$ and also calculated by use of lattice parameters, and melting points obtained by use of DTA curves, which were shown in Fig.3, at a heating rate of $10^\circ\text{C}/\text{min}$. The observed densities almost agreed with calculated ones. The density of $\text{LiNd}(\text{WO}_4)_2$ crystals was nearly equal to that of $\text{NaNd}(\text{WO}_4)_2$ crystals. This reason seems to be that the increasing ratio¹⁸⁾ of unit cell volume of $\text{NaNd}(\text{WO}_4)_2$ crystals is nearly equal to that²⁰⁾ of formula weight(see Table 2). On the other hand, the density of $\text{KNd}(\text{WO}_4)_2$ crystals was smaller than those of the former two crystals. This reason seems to be that the increasing ratio¹⁹⁾ of unit cell volume of $\text{KNd}(\text{WO}_4)_2$ crystals is larger than that²¹⁾ of formula weight(see Table 2).

On the basis of the values of densities, the numbers of Nd ion in $\text{RNd}(\text{WO}_4)_2$ crystals were approximately estimated to be $6.3 \times 10^{21} \text{ cm}^{-3}$ for Li, $6.1 \times 10^{21} \text{ cm}^{-3}$ for Na and $5.8 \times 10^{21} \text{ cm}^{-3}$ for K, respectively.

In these crystals, $\text{NaNd}(\text{WO}_4)_2$ crystals had the highest melting point of $1235\pm 5^\circ\text{C}$.

Table 3 Densities and melting points of $\text{RNd}(\text{WO}_4)_2$ crystals

Crystal	$d_{\text{obs.}}$ (g/cm^3)	$d_{\text{cal.}}$ (g/cm^3)	mp ($^\circ\text{C}$)
$\text{LiNd}(\text{WO}_4)_2$	6.83 \pm 0.02	6.79	1130 \pm 5
$\text{NaNd}(\text{WO}_4)_2$	6.78 \pm 0.02	6.80	1235 \pm 5
$\text{KNd}(\text{WO}_4)_2$	6.55 \pm 0.02	6.55	1085 \pm 5

Fig.3 DTA curves of grown $\text{RNd}(\text{WO}_4)_2$ crystals

It seems probable that the thermal stabilities of $RNd(WO_4)_2$ crystals are generally changed by both the radii of R ions and their polarizing powers on anionic groups (WO_4^{2-}) when R ions are substituted with one another in the crystal constituents.

So far as these properties (brittleness, melting point and transparency) examined above are concerned, $NaNd(WO_4)_2$ crystals may be one of the best ceramic material in the three kinds of $RNd(WO_4)_2$ crystals.

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- 18) The increasing ratio (1.023) is given by [unit cell volume of $NaNd(WO_4)_2$] / [unit cell volume of $LiNd(WO_4)_2$].
- 19) The increasing ratio (1.088) is given by [unit cell volume of $KNd(WO_4)_2$] / [unit cell volume of $LiNd(WO_4)_2$].
- 20) The increasing ratio (1.025) is given by [formula weight of $NaNd(WO_4)_2$] / [formula weight of $LiNd(WO_4)_2$].
- 21) The increasing ratio (1.050) is given by [formula weight of $KNd(WO_4)_2$] / [formula weight of $LiNd(WO_4)_2$].

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